

1 **Heteroatom-doped graphene aerogels and carbon-magnetite catalysts for the**
2 **heterogeneous electro-Fenton degradation of acetaminophen in aqueous solution**

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26 **Abstract**

27 Nitrogen (N) and nitrogen/sulphur (N/S) co-doped graphene aerogels were synthesized by
28 hydrothermal method using graphene oxide and urea or thiourea as precursors following a
29 freezing-drying process. The determination of the number of the electrons involved in the
30 mechanism of electro-reduction of oxygen (ORR) in alkaline medium was assessed. The
31 mechanism of $4e^-$ was obtained with N/S-doped graphene aerogels showing good
32 electrocatalytic behavior and larger kinetic current densities. The $2e^-$ ORR process (involving
33 in situ H_2O_2 generation) was mainly obtained using N-doped graphene aerogels and it was
34 related to the content of N functionalities as well as to the porous texture of the samples.
35 Carbon-magnetic catalyst was synthesized to promote the decomposition of H_2O_2 to HO^\bullet for
36 the degradation of acetaminophen (ACE) in heterogeneous electro-Fenton (EF) process. The
37 highest pollutant removal (71.0%) and mineralization (51.6%) were obtained with N-doped
38 materials and carbon- Fe_3O_4 catalyst, showing good stability and low iron leaching (0.33 mg L^{-1}).
39 ¹).

40 **Keywords:** *Graphene; Electro-reduction of oxygen; Carbon-magnetite; Heterogeneous*
41 *electro-Fenton; Nitrogen and sulphur doping.*

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51 **1. Introduction**

52 Water pollution is an immediate consequence of population growth, agricultural and industrial
53 expansion and is on the limit of a critical point in several zones of the globe. Pharmaceutical
54 compounds have been found not only in the sea but also in tap water [1], with the direct damage
55 in worldwide public health [2, 3]. Acetaminophen (ACE, also known as paracetamol) whose
56 chemical name is *N*-acetyl-*p*-aminophenol, is one of the most present drugs found on waste
57 water [1, 4, 5] due to aspects as poor removal efficiency [6] or toxic impact on the environment
58 [7]. ACE is one of the most commonly used analgesics with and without prescription and has a
59 great impact in the health causing liver injury, hepatocyte necrosis and even mortality. Among
60 58-68% of the molecule is excreted from the body and finishes in municipal waste system [8].
61 In the last few decades, advanced oxidation processes (AOPs), including chemistry,
62 photochemistry, photocatalytic, electrochemistry and photo-electrochemistry methods have
63 been developed as powerful and environmentally-friendly technologies for the complete
64 decomposition of organic contaminants [9]. AOPs consist on the generation of highly reactive
65 species, mainly hydroxyl radicals (HO[•]) [10, 11] being possible to obtain the complete
66 mineralization of organic pollutants into H₂O, CO₂ and inorganic ions. Among the AOPs, the
67 Fenton-based process has become an ideal method because of its high performance and easy
68 operation. The classical Fenton process involves the production of HO[•] radicals from hydrogen
69 peroxide (H₂O₂) and Fe²⁺ or Fe³⁺ ions at acidic pH values (2.5-3.0) [12], following the Eq. 1
70 [13]:



72 One of the ecofriendly and more cost-effective versions of this kind of process, is the electro-
73 Fenton (EF) process. This technology is based on the electrochemical generation of HO[•]
74 radicals through (i) the *in situ* electrogeneration of H₂O₂ at the cathode by oxygen reduction

75 reaction (ORR) which mainly consists of $2e^-$ process (Eq. 2) instead $4e^-$ process that generate
76 H_2O (Eq. 3); (ii) the addition of Fe^{2+} catalyst to the solution and (iii) the continuous regeneration
77 of Fe^{3+} to Fe^{2+} at the cathode [14].



80 It well known that various carbon-based materials such as carbon fiber [15], carbon black [16],
81 carbon nanotubes [17] and graphene [18], among others, have been suggested interesting
82 alternatives to precious metals as cathode for EF process because of their nontoxicity, excellent
83 conductivity, good stability and low cost. Graphene derivatives (e.g., graphene oxide, GO and
84 reduced graphene oxide, rGO) and more recently three-dimensional (3D) graphene-based
85 aerogel [19] have been investigated as ORR catalysts in alkaline electrolytes due to its large
86 specific surface area, high electrical conductivity, mechanical strength, easy functionalization
87 [20] and multiplex electron transfer pathways [21]. It is noteworthy that the presence of
88 heteroatoms atoms such as O, N, B, P, and S, as well as defect sites on the lattice of graphene
89 would influence the adsorption of O_2 , enhancing ORR activity [22-25].

90 Heterogeneous EF process normally uses solid catalyst source such iron species (or transition
91 metals) supported on a porous matrix for the decomposition of H_2O_2 to HO^\bullet radicals. In this
92 regard, several materials have been reported as catalyst supports, i.e. ionic exchange resins,
93 silicates and carbon-based materials, among others [26]. Activated carbon is one of the most
94 interesting catalyst support due to its low cost as well as textural, chemical and morphological
95 tunable properties, which can play a significant role during the degradation of organic pollutants
96 [27, 28].

97 Heterogeneous mode unravel the problem associated with the generation of sludge, which
98 should be eliminated by costly processes [29]. In this sense, the utilization of magnetic catalyst
99 has been reported as an easier and efficient way of catalyst recovery under a magnetic field,

100 showing reusability properties [30]. Although the use of iron supported on activated carbon has
101 been previously reported as catalyst in EF reaction for the degradation of several pollutants,
102 such as phenol [28] or methyl blue [31], to the best of our knowledge there are no works in
103 literature using both heteroatom-doped graphene aerogels and magnetic carbon-iron oxide
104 particles in the heterogeneous electro-Fenton degradation of acetaminophen in aqueous
105 solution.

106 In view of these precedents, the aim of this work is to study the potential of N- and N/S-doped
107 graphene aerogels as cathode in oxygen reduction reaction and evaluate the electrons involved
108 in the process; i.e. 4 or 2 electrons for H₂O or H₂O₂ generation, respectively. The effect of N
109 or N/S species on ORR activity will be also discussed. Carbon-Fe₃O₄ catalysts are used to
110 promote the decomposition of H₂O₂ to HO• radicals for the sake of improving both conversion
111 and mineralization of ACE in heterogeneous electro-Fenton process.

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113 2. Materials and methods

114 2.1. Chemicals

115 Graphite powder < 20 μm, acetaminophen, ACE (C₈H₉NO₂, Mr 151,163 g mol⁻¹, 99.5 wt.%
116 [CAS number: 103-90-2]) and formaldehyde (99.5%) were purchased from Sigma-Aldrich.
117 Sulphuric acid (H₂SO₄, 95-98%) and potassium permanganate (KMnO₄, 99.0%) were obtained
118 from PanReac AppliChem. Sodium nitrate (NaNO₃, >99.0%), urea (NH₂CONH₂, 99.5%) and
119 thiourea (CH₄N₂S, 99%) from Alfa Aesar. *Tert*-butanol (C₄H₁₀O) was obtained from Fluka.

120

121 2.2. Synthesis of graphene oxide (GO)

122 GO was synthesized from synthetic graphite (powder, particle size < 20 μm) by a modified
123 Hummers method [32, 33] described elsewhere [33]. The resultant material was dispersed in
124 distilled water (2g L⁻¹) and sonicated in an ultrasound bath for one hour. After that, the
125 unexfoliated graphite oxide was removed by centrifugation (20 min at 3000 rpm).

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128 *2.3.Synthesis of N and S-doped graphene aerogels*

129 N-doped graphene and N/S-doped graphene aerogels were obtained by hydrothermal reduction
130 of GO suspension (1.3 g/L) with urea [34] or thiourea as N or N/S precursor, respectively. In a
131 typical procedure, the appropriate amount of urea or thiourea was dissolved in 60 mL of GO
132 suspension and transferred into a 100 mL Teflon autoclave and subjected to hydrothermal
133 treatment at 180°C for 12h. The as-prepared resultant sponge was immersed in distilled water
134 for 4 days to remove residues. Then, the graphene-based material was placed in acetone for 3
135 days making an acetone changing twice a day. Subsequently, the solid was filtered and placed
136 on tert-butanol and finally the solvent was eliminated by freeze drying. The materials were
137 labelled as rGO-N-X or rGO-N/S-X when using urea or thiourea as precursors, respectively
138 where X refers to the ratio GO:precursor (i.e. 5, 10 and 20 for 1:5, 1:10 and 1:20 ratio of
139 GO:urea or GO:thiourea, respectively). For the comparison purpose, the rGO aerogel was also
140 synthesized via the same procedure without the addition of urea or thiourea.

141

142 *2.4.Synthesis of magnetite and carbon-magnetite*

143 The iron oxide (Fe_3O_4) magnetic particles were synthesized by the method previously described
144 by Petcharoen and Sirivar [35]. In a typical procedure, 1.5g of FeCl_2 and 3g of FeCl_3 (molar
145 relation of 2; $\text{Fe}^{3+}/\text{Fe}^{2+}$ in solution) were dissolved in 100 mL of distilled water under N_2 flow
146 with vigorous stirring, then 10 mL of NH_4OH at 25% was added to the resultant mixture
147 producing a color change from orange to black. The magnetic particles precipitated were
148 washed with deionized water, ethanol and dried at 80°C for 8 hours. The iron oxide particles
149 will be denoted as Fe_3O_4 .

150 Carbon-magnetite composite was prepared using a mix of resorcinol-formaldehyde as the
151 carbon source. In a typical procedure, a solution of the obtained Fe_3O_4 in 50 mL of NH_4OH

152 (30%) was added to 500 mL of n-heptane with a stirring of 450 rpm and heated at 70°C. The
153 amount of magnetite was calculated according 10% of final weight of carbon gel. A solution of
154 resorcinol (R), formaldehyde (F) and water (W) in proportion R/F=1 and R/W=2 (containing
155 25g of R, 45 mL of F and 58 mL of W), was added dropwise to de Fe₃O₄/n-heptane solution.
156 The resultant gel was aged at 70°C for 24 h under stirring. After that, the suspension formed
157 was filtered and the solid placed in acetone for 2 days [36] making an acetone changing twice
158 a day in order to reduce porosity collapse with the following drying step [37]. After 2 days in
159 acetone the solid was filtered and placed on tert-butanol, the solvent was eliminated by freeze
160 drying. The obtained material was treated in a furnace with N₂ flow at 2 °C min⁻¹ until 900°C
161 and with soak time of 2 h. The carbon-magnetite composite is referred as carbon-Fe₃O₄.

162

163 *2.5.Characterization techniques*

164 Textural characterization was carried out by N₂ adsorption at -196 °C, using a Quantachrome
165 Autosorb-1 equipment. The BET and Dubinin–Radushkevich equations were applied to
166 determine the apparent surface area (S_{BET}) and the micropore volume (W₀) and the mean
167 micropore width (L₀), respectively. Pore size distributions were obtained by applying the
168 Barret-Joyner-Hallender (BJH) method. The mesopore volume of the samples (V_{mes}) were
169 obtained from the volume of N₂ adsorbed at P/P₀ = 0.95 and the micropore volume (W₀). The
170 chemical characterization of the composites was further analyzed by X-ray photoelectron
171 spectroscopy (XPS). The spectra were obtained on a Kratos Axis Ultra-DLD X-ray
172 photoelectron spectrometer equipped with a hemispherical electron analyzer operating at 12 kV
173 and 10 mA connected to a detector DLD (delay-line detector). Raman spectra were obtained
174 using a Micro-Raman JASCO NRS-5100 dispersive spectrometer with a 532 nm laser line.
175 Attenuated total reflection Fourier transform infrared (ATR-FTIR) spectra were recorded on a
176 NICOLET 510P spectrometer using ZeSn as ATR crystal. The total oxygen content of samples
177 was obtained with an elemental CHNS-O Analyzer Flask (1112 Series) from Thermo Finigan.

178 The morphology of the materials was analyzed by scanning electron microscopy (SEM) using
179 a LEO (Carl Zeiss) GEMINI-1530 microscope. Transmission electron microscopy (TEM) was
180 performed in a LIBRA 120 PLUS (Carl Zeiss) equipment. The iron oxide phase and
181 crystallinity were determined by a powder X-ray diffraction (XRD) pattern using a Bruker D8
182 Advance X-ray diffractometer equipped with Cu K α radiation at a wavelength (λ) of 1.541 Å.

183

184 *2.6. Electrochemical characterization*

185 To test the oxygen reduction activity of the graphene-based aerogels, cyclic voltammetry (CV)
186 and linear sweep voltammetry (LSV) experiments were carried out in a three-electrode cell
187 controlled by a Biologic VMP multichannel potentiostat (BioLogic Science Instruments;
188 France). Both Ag/AgCl and Pt-wire were selected as reference and counter electrodes,
189 respectively. The three electrodes were immersed in a 0.1 M KOH water solution. As a working
190 electrode, a Rotating Disk Electrode (RDE) was used, i.e. Metrohm AUTOLAB RDE-2 with a
191 3 mm Glassy Carbon tip (Metrohm AG; Switzerland). The working electrode was prepared by
192 the deposition on the RDE tip of 10 μ l of a suspension containing 5 mg of the graphene-based
193 materials dispersed in 1 mL of a Nafion (5%) and water (1:9, v:v ratio) solution sonicated for
194 30 minutes.

195 CV experiments were carried out while bubbling N₂ or O₂ through the electrolyte solution
196 during the measurements with the potential interval of -0.8 to 0.4 V at a scan rate of 5 mVs⁻¹
197 and 50 mV s⁻¹. LSV curves were performed in a 0.1 M KOH solution saturated with O₂ at
198 different rotation speeds (i.e. 500, 1000, 1500, 2000, 2500, 3000 and 4000 rpm) with a scan
199 rate 5 mVs⁻¹ and sweeping voltage ranged from 0.4 to -0.8 V. The experimental data were fitted
200 to the Koutecky-Levich model (Eqs. 4 and 5) in order to assess the electrocatalytic performance
201 of samples and to calculate the number of electrons transferred for each of them.

$$202 \quad \frac{1}{J} = \frac{1}{J_k} + \frac{1}{B\omega^{0.5}} \quad (4)$$

$$B = 0.2nF(D_{O_2})^{2/3}\nu^{-1/6}C_{O_2} \quad (5)$$

where j is the current density; J_k is the kinetic current density; ω correspond to the rotation speed; F is the Faraday constant; ν is the viscosity ($0.01 \text{ cm}^2 \text{ s}^{-1}$) and D_{O_2} and C_{O_2} are the oxygen diffusion coefficient ($1.9 \times 10^{-5} \text{ cm}^2 \text{ s}^{-1}$) and oxygen concentration ($1.2 \times 10^{-6} \text{ mol cm}^{-3}$), respectively [38, 39].

208

209 *2.7. Adsorption kinetics and isotherms*

210 ACE adsorption kinetics were studied in order to establish the time required to achieve the
211 adsorption equilibrium at saturation, reaching the equilibrium in 72 h (results not shown).

212 Adsorption isotherms of ACE from aqueous solutions were obtained by using 0.01 g of
213 prepared materials and 25 mL of pharmaceutical solutions at concentrations ranging from 10 to

214 100 mg L^{-1} . The suspensions were mechanically shaken until equilibrium was reached (72 h)

215 and measurements were performed at least twice to know their reproducibility. Equilibrium

216 concentrations were determined at 242 nm using a UV-spectrophotometer model UV-1800

217 Shimadzu. Adsorption isotherms were fitted to Langmuir and Freundlich equations (Eqs. 6 and

218 7, respectively):

$$219 \quad \frac{C}{X} = \frac{1}{BX_m} + \frac{C}{X_m} \quad (6)$$

$$220 \quad X = K_F \times C^{1/n} \quad (7)$$

221 where X is the amount adsorbed at a given equilibrium concentration, C , X_m is the adsorption

222 capacity, and B is the Langmuir constant. In the Freundlich equation C and X have the same

223 meaning, n (dimensionless) is an empirical parameter, and K_F is the so-called unit capacity

224 factor. So, K_F would be equivalent to the amount adsorbed at an equilibrium concentration of

225 1 mg L^{-1} .

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229 *2.8. Electro-Fenton experiments*

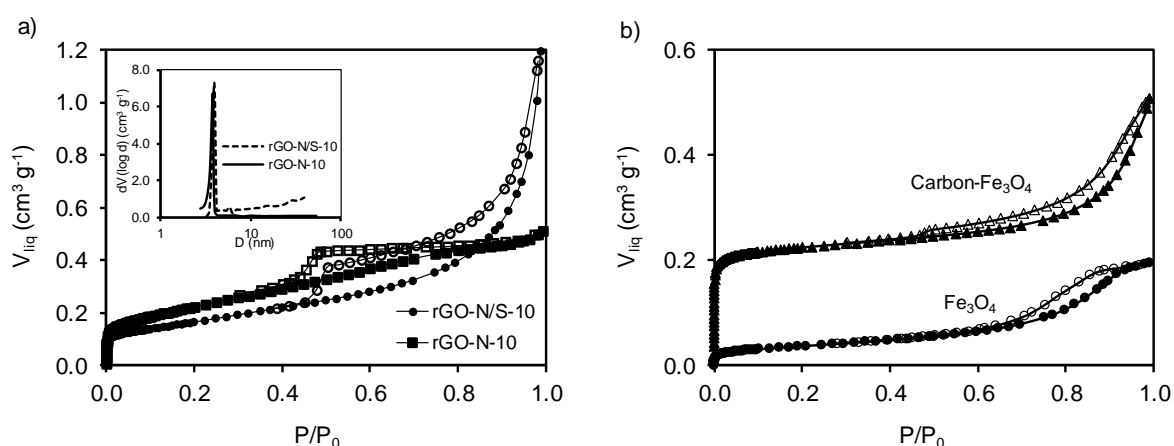
230 The electro-Fenton (EF) experiments were carried out in an undivided glass cell with three
231 electrodes controlled by a Biologic VMP multichannel potentiostat (BioLogic Science
232 Instruments), containing 120 mL of ACE solution vigorously stirred with a magnetic bar. The
233 experiment setup of the device was as potentiostatic mode at -0.5V and -0.55V, based on the
234 ORR results. The reference electrode used was Ag/AgCl and the counter electrode was Pt-wire.
235 The working electrode (cathode) was prepared as previously reported by our group [40]. A
236 paste of 50 mg of graphene-based aerogels and polytetrafluoroethylene, PTFE (60% suspension
237 in water) was homogenous mixed with a ratio of 90:10. After that, the mixture was dried at
238 80°C overnight and the resulting material was homogeneously loaded on both sides of a
239 graphite sheet (50 mm x 8 mm). Solution of 10 mg L⁻¹ of ACE and 0.5 M Na₂SO₄ as electrolyte
240 were used. The pH value of electrolyte was adjusted at 3.0 by H₂SO₄. Carbon-Fe₃O₄ (0.25 g L⁻¹)
241 ¹) was added as catalysts. The O₂ gas was bubbled for 30 minutes before the experiment started
242 in order to saturate the solution and continuously bubbled during the experiment time. As
243 reference ACE removal without the addition of Fe₃O₄ or carbon- Fe₃O₄ was also conducted.
244 Samples for analysis were taken periodically from the glass cell, with immediate addition of a
245 proper amount of Na₂SO₃ for H₂O₂ reduction. The concentration of ACE was monitored using
246 a UV-spectrophotometer model UV-1800 Shimadzu at 242 nm. The concentration of dissolved
247 iron was analyzed colorimetrically by measuring the absorbance at 510 nm, according to ISO
248 6332:1988. The total organic carbon (TOC), determined with a TOC-VCSH Shimadzu
249 equipment, was used to evaluate the mineralization.

250

251 **3. Results and discussion**

252 *3.1. Characterization of the graphene-based materials and magnetic catalysts*

253 The porous texture of all the graphene-based materials and magnetic catalysts was studied by
 254 N₂ adsorption-desorption and the results are given in Table 1. N₂ adsorption-desorption
 255 isotherms for rGO-N-10 and rGO-N/S-10 and Fe₃O₄ and carbon-Fe₃O₄ catalysts are shown in
 256 Fig. 1a-b, respectively. All isotherms of graphene-based materials can be described as type-II,
 257 in accordance with IUPAC classification [41]. The progressive increase of the N₂ adsorbed
 258 volume to higher relative pressures and the formation of a hysteresis loop of type H3 in both
 259 samples, denotes a strong influence of mesoporosity on the isotherm profile.



260
 261 **Fig. 1.** a) Nitrogen adsorption-desorption isotherms and pore size distributions (inset; BJH
 262 method) of rGO-N-10 and rGO-N/S-10; b) nitrogen adsorption-desorption isotherms of Fe₃O₄
 263 and Carbon-Fe₃O₄.

264
 265 The low surface area observed for GO (21 m² g⁻¹) seems to be related with the re-stacking of
 266 the graphene sheets when the suspension is dried as previously observed [42]. It is shown that
 267 a large increase in the S_{BET} and micropore and mesopore volume were observed for both N-
 268 and N/S-doped samples in comparison with GO (Table 1). These results are likely due to the
 269 partial exfoliation of GO during the preparation of the aerogels. It is noteworthy that the S_{BET},
 270 micropore volume (W₀) of N-doped graphene aerogels were significantly higher than those
 271 obtained for N/S- doped graphene aerogels (Table 1). These values increased with the content
 272 of N, obtaining the highest S_{BET} and micropore volume for rGO-N-20 (592 m² g⁻¹ and 0.229 cm³

273 g⁻¹, respectively). Moreover, comparing the N₂ adsorption-desorption isotherms for both rGO-
 274 N-10 and rGO-N/S-10 (Fig. 1a), a progressive increase of the N₂ adsorbed volume to higher
 275 relative pressures (indicative of adsorption in wide micropores and mesopores) was observed.
 276 On the other hand, mesopores volume of N/S-doped graphene aerogels were higher compared
 277 to those in N-doped graphene aerogels (Table 1). The pore size distributions of N- and N/S-
 278 doped graphene aerogels indicated mesopore diameter smaller than 6.0 nm. Such an improved
 279 surface area and pore size distribution in the doped graphene aerogels could shorten the
 280 diffusion path and provide more channels that may facilitate more efficiently the electrolyte
 281 diffusion through the mesopores.

282

283 **Table 1.** Textural properties of the graphene-based materials and magnetic catalysts

284

Sample	S _{BET} (m ² g ⁻¹)	W ₀ (cm ³ g ⁻¹)	L ₀ (nm)	V _{mes} (cm ³ g ⁻¹)
rGO-N-5	583	0.228	1.44	0.26
rGO-N-10	489	0.191	1.51	0.28
rGO-N-20	592	0.229	1.41	0.30
rGO-N/S-5	441	0.178	1.35	0.55
rGO-N/S-10	374	0.143	1.46	0.55
rGO-N/S-20	432	0.164	1.46	0.57
GO	21	--	--	--
Fe ₃ O ₄	78	--	--	0.18
Carbon-Fe ₃ O ₄	551	0.216	0.68	0.18

293

294 Regarding magnetic catalysts (i.e. Fe₃O₄ and carbon-Fe₃O₄, Fig. 1b), the adsorption isotherms
 295 belong to type IV with the H4 hysteresis loop. The specific surface area of the magnetic
 296 catalysts increased with the presence of carbon gel (i.e. 78 to 551 m² g⁻¹ for Fe₃O₄ and carbon-
 297 Fe₃O₄, respectively). Moreover, the results show that carbon-Fe₃O₄ presents an important micro

298 and mesopores network pore volumes: 0.216 and 0.18 cm³ g⁻¹, respectively and a micropore
299 width (L_0) of 0.68 nm.

300 Table 2 listed the results of the elemental analysis of the graphene-based materials showing the
301 larger percentage of oxygen content for GO (i.e. 53.5%). It is interesting to note that mostly of
302 oxygenated groups were removed during the thermal treatment and doping of the materials
303 showing a N or N/S incorporation (depending of the precursor, i.e. urea or thiourea,
304 respectively) at the same time.

305

306 **Table 2.** Elemental analysis (% wt.) and intensity ratios of the D bands relative to the G mode
307 for the graphene-based materials.

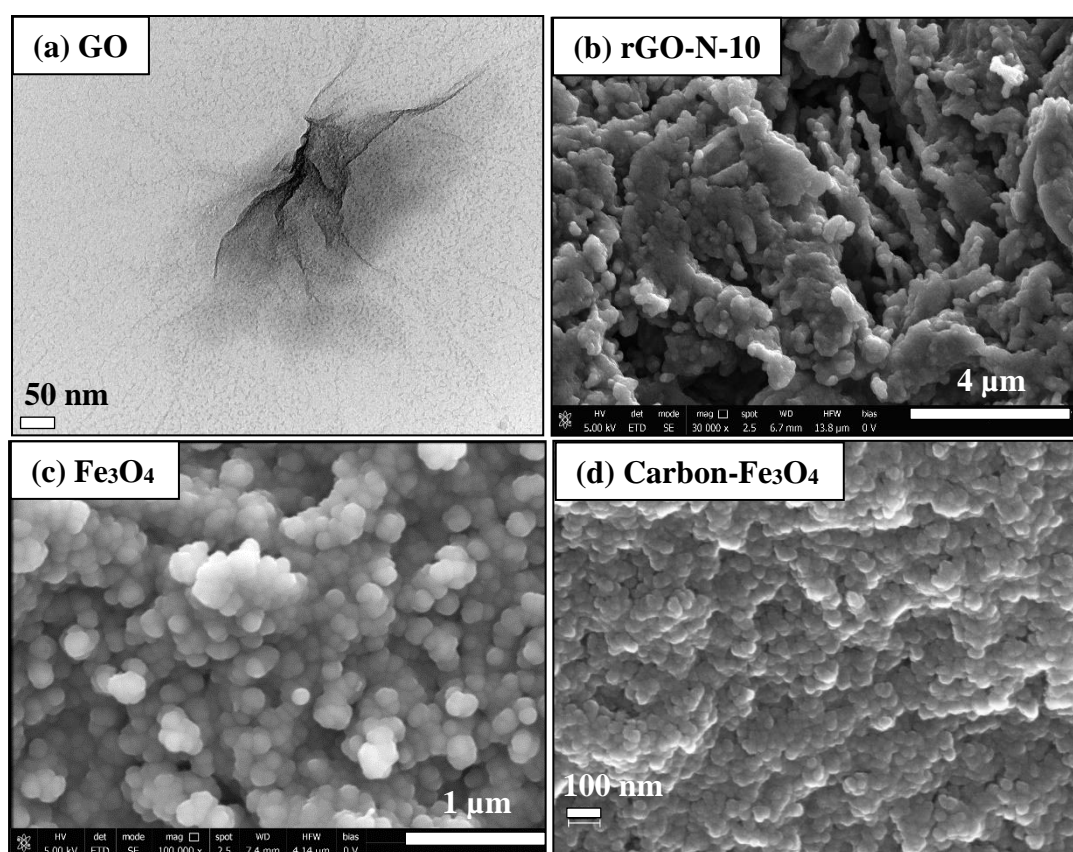
Sample	N	C	H	S	O	I_D/I_G
	%	%	%	%	%	532 nm
rGO-N-5	7.1	69.3	3.1	0.0	20.5	1.11
rGO-N-10	8.2	70.7	3.0	0.0	18.1	1.08
rGO-N-20	9.3	70.2	2.8	0.0	17.7	1.09
rGO-N/S-5	1.7	71.3	3.0	1.1	22.9	1.13
rGO-N/S-10	1.8	72.1	3.4	1.3	21.4	1.16
rGO-N/S-20	2.1	72.8	3.7	1.4	20.2	1.12
rGO	0.0	71.5	5.2	0.1	23.2	1.07
GO	0.1	42.6	3.6	0.2	53.5	0.85

308

309 In general, as the ratio of GO:urea or GO:thiourea increased, the N and S content in the samples
310 also increased (Table 2). In particular, 9.3% of N was determined for rGO-N-20 whereas the
311 material prepared with the highest ratio GO:thiourea, (i.e. 1:20 in rGO-N/S-20) showed values
312 of 2.1% and 1.4% of N and S, respectively. It seems that N-doping was more effective than N/S
313 doping, in agreement with previous publications [43, 44].

314 TEM and SEM analysis were performed for the visualization of the morphology of the prepared
315 materials. TEM micrograph of the GO is shown in Fig. 2a, indicating the presence of graphene

316 multilayers. Typical SEM micrographs of rGO-N-10 (Fig. 2b), Fe₃O₄ (Fig. 2c) and carbon-
317 Fe₃O₄ (Fig. 2d) clearly reveal the differences in the morphology of the materials. rGO-N-10
318 exhibit 3D interconnected structure of aggregated wrinkled graphene layers. SEM image of
319 Fe₃O₄ (Fig. 2c) indicated clusters of spherical-like particles whereas the micrograph of carbon-
320 Fe₃O₄ showed a more compact structure which can be explained by the introduction of
321 magnetite in the crystalline space of the gel (Fig. 2d).



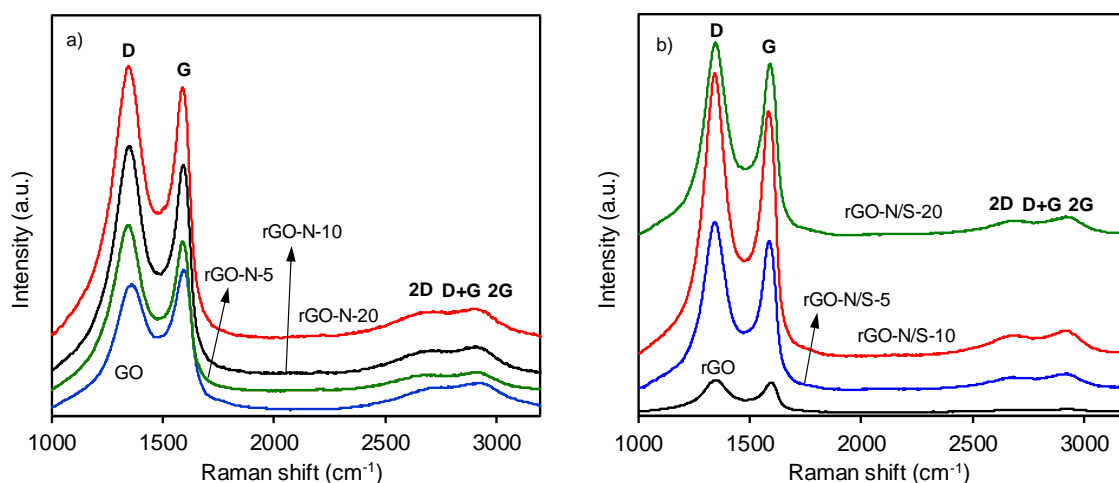
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335 **Fig. 2.** a) TEM micrograph of GO and SEM micrographs of b) rGO-N-10; c) Fe₃O₄ and d)
336 Carbon-Fe₃O₄.

337

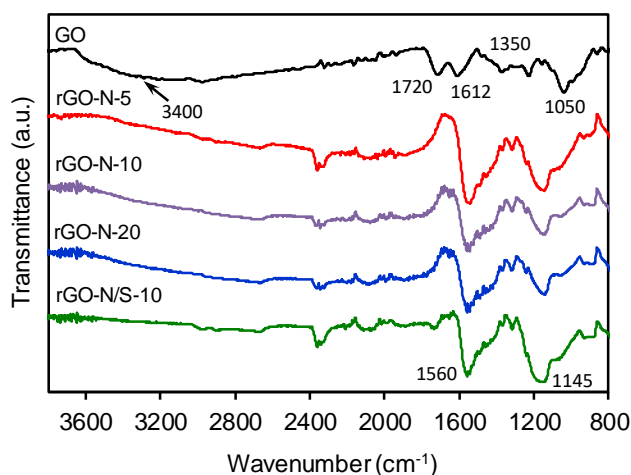
338 Raman spectra are shown in Fig. 3. All samples exhibited the graphitic G band at ~1582 cm⁻¹,
339 D band ~1345 cm⁻¹ and the high frequency modes, including the 2D band (~2700 cm⁻¹) that
340 result of an overtone of the D band, as well a D+G peak which is result of a combination of D
341 and G peaks and appears over 2950 cm⁻¹ [45]. The graphitic G band corresponds to the

342 stretching vibration mode of sp^2 carbon atoms [46, 47], the D-band reflect the disorder-induced
 343 character in the aromatic rings whereas its overtone (2D-band) give information in sp^3
 344 generated defects (D-band) and exfoliation of individual graphene sheets (2D band) that lead
 345 to electronic decoupling [48]. Defects in graphitic plane are important due to the high
 346 anisotropy in the mechanical strength and the in/out plane electrical conductivity in order to
 347 improve the performance of carbon based materials in different applications [49].



348
 349 **Fig. 3.** Raman spectra of a) GO and N doped graphene materials and b) rGO and N/S doped
 350 graphene materials.

351
 352 Table 2 listed the Raman intensity ratios of the D bands relative to the G mode (I_D/I_G) obtained
 353 for the graphene-based materials, calculated from the areas of the corresponding Raman bands,
 354 at 532 nm. The I_D/I_G ratio is a degree representation of defects and disorder levels of structural
 355 ordered carbon in the graphitic materials [50]. As seen in Table 2, the I_D/I_G ratio was larger for
 356 rGO and for the heteroatom doped graphene materials comparing to GO. This change can be
 357 attributed to a decrease in the average size of the sp^2 domains upon removal of oxygen groups
 358 from GO during the doping process, suggesting the presence of abundant defects in the
 359 materials [51]. In particular, both rGO-N-10 and rGO-N-20 presented the lowest I_D/I_G values
 360 (i.e. 1.08 and 1.09, respectively) suggesting that these materials contain the lowest amount of
 361 defects.



362

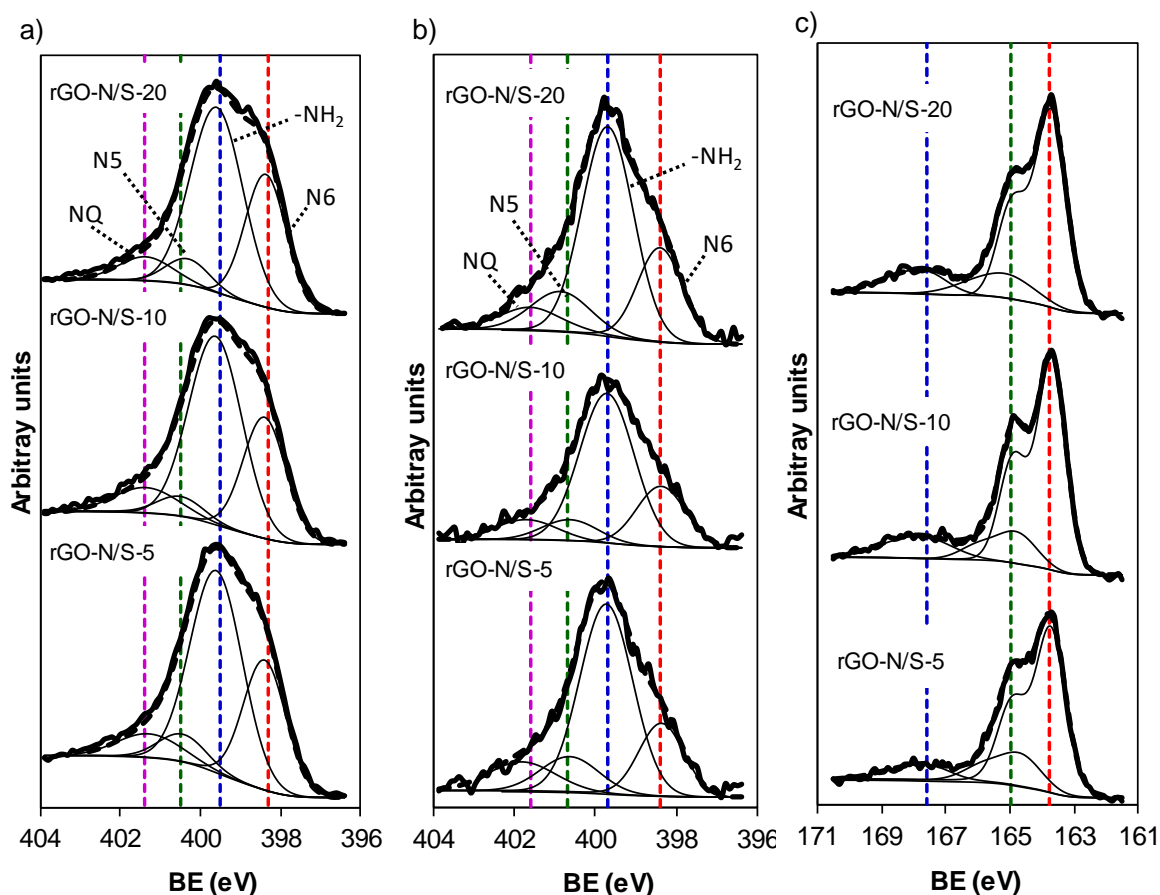
363 **Fig. 4.** FTIR-ATR spectra of GO, rGO-N-5, rGO-N-10, rGO-N-20 and rGO-N/S-10.

364

365 FTIR-ATR were studied for all the materials. In particular, the spectra of GO, rGO-N-5, rGO-
 366 N-10, rGO-N-20 and rGO-N/S-10 are depicted in Fig. 4. GO showed a broad band situated ca.
 367 3400 cm^{-1} , assigned to the vibration of C-OH groups, a band at ca. 1720 cm^{-1} attributed to
 368 carbonyl groups, C=O, and a band at ca. 1350 cm^{-1} corresponds to C-OH stretching. The band
 369 at ca. 1220 cm^{-1} corresponds to breathing vibrations of epoxy groups (-O-) and the band at ca.
 370 1050 cm^{-1} is attributed to the stretching vibration of C-O groups. The band at ca. 1612 cm^{-1} can
 371 be assigned to the skeletal vibration of graphene sheets. Compared to that of GO, the peak
 372 intensities of the oxygen functional groups for the heteroatom graphene-based materials
 373 become very weak indicating the partial removal of the oxygen-containing groups during the
 374 doping process [42]. For rGO-N-5, rGO-N-10, rGO-N-20 and rGO-N/S-10 materials, a band
 375 likely corresponding to sp^2 aromatic C=C and C=N bonds [52] was observed at ca. 1560 cm^{-1} .
 376 Moreover, the rGO-N/S-10 showed a more prominent peak at around 1100-1145 cm^{-1} which
 377 most probably corresponds to the peak of the C-S-C bond symmetrical stretching vibration [53],
 378 but also possible to the other bonds of C-O-C, that explain the peak in the samples without
 379 sulphur doping such as rGO-N-5, rGO-N-10 and rGO-N-20.

380

381



382
 383 **Fig. 5.** a) N1s deconvoluted spectra for rGO-N-5, rGO-N-10 and rGO-N-20; b) and c) N1s and
 384 S2p XPS deconvoluted spectra for rGO-N/S-5, rGO-N/S-10 and rGO-N/S-20, respectively.

385
 386 To further investigate the nitrogen and sulphur content on the heteroatom graphene samples,
 387 X-ray photoelectron spectroscopy (XPS) signals were analyzed as illustrated in Fig. 5. XPS
 388 data for nitrogen, sulphur and oxygen contents are collected in Table 3. Fig. 5a shows the
 389 deconvoluted N1s XPS spectra for rGO-N-5, rGO-N-10 and rGO-N-20, revealing the presence
 390 of four types of species, namely graphitic N (or quaternary N, NQ), pyrrolic N (N5), amino
 391 groups (-NH₂) and pyridinic N (N6) at ~401.3, ~400.4, ~399.6 and ~398.4 eV, respectively,
 392 consistent with the results for other N-doped carbon materials [54, 55]. It is interesting to
 393 mention that the most abundant N species in the N-doped graphene materials correspond to the
 394 amino groups (-NH₂), followed by the N-pyridinic species (N6) and finally both N-pyrrolic
 395 form (N5) and N-graphitic (NQ) with the lowest amount as previously observed [42]. It can be

396 generally accepted that the covalent functionalization with amino groups can occur at the edge
 397 or defect sites of GO, where are more chemically active than those within the plane of graphene
 398 [56].

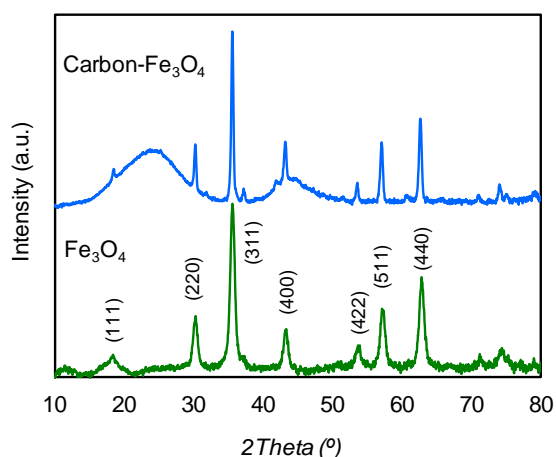
399
 400 **Table 3.** Nitrogen and sulphur species percentage and corresponding binding energies (in
 401 brackets, eV) and nitrogen, sulphur and oxygen content (% wt.) obtained by XPS analysis.

Sample	N (%)	S (%)	O (%)	N Peak (%)				S Peak (%)		
				(N6)	(-NH ₂)	(N5)	(NQ)	C-S-C	R ₂ -S=O	C-SO _x -C
rGO-N-5	7.6	--	11.6	29 (398.4)	56 (399.6)	7 (400.4)	8 (401.3)	-	-	-
rGO-N-10	7.6	--	12.8	29 (398.4)	57 (399.6)	5 (400.5)	9 (401.4)	-	-	-
rGO-N-20	7.8	--	11.0	31 (398.4)	55 (399.6)	7 (400.3)	8 (401.4)	-	-	-
rGO-N/S-5	1.8	2.0	17.0	19 (398.4)	58 (399.8)	12 (400.7)	11 (401.8)	73 (163.7)	18 (164.6)	9 (167.5)
rGO-N/S-10	1.5	2.1	15.7	22 (398.4)	60 (399.7)	9 (400.7)	10 (401.7)	73 (163.7)	14 (164.7)	13 (167.6)
rGO-N/S-20	1.9	1.9	15.9	22 (398.4)	57 (399.7)	13 (400.9)	8 (401.6)	43 (163.7)	14 (165.0)	12 (167.6)

402
 403 The N/S doped graphene materials, N1s and S2p XPS deconvoluted spectra are shown in Figs.
 404 5b and c, respectively. The N1s XPS spectra for rGO-N/S-5, rGO-N/S-10 and rGO-N/S-20
 405 showed also the presence of pyridinic-N, amino, pyrrolic-N and graphitic-N functionalities
 406 (Table 3). Regarding the deconvoluted S2p XPS spectra, each S2p peak is the resultant of two
 407 components (2p_{1/2} and 2p_{3/2}) although only the sum of both peaks was plotted in Fig. 5c, being
 408 2p_{3/2} binding energy the one used to assign different peaks to chemical functionalities.
 409 Considering that, the deconvoluted S2p XPS spectra showed three different **chemically shifted**
 410 **components**, which are attributed to the S-thiophene-like groups (C-S-C) at ~163.7 eV,
 411 sulphoxide groups (R₂-S=O) at ~164.6 eV and a low intense **component** associated with

412 oxidized forms of sulphur, i.e. sulphone groups (C-SO_x-C) at ~167.5 eV, as previously reported
413 [57, 58]. Once again, the predominant nitrogen specie in the N/S- doped-graphene materials
414 was the amino group (-NH₂), whereas the most abundant sulphur-containing functionalities
415 correspond to the S-thiophene (Table 3) consistent with previous results in literature for
416 graphene-doped materials [42].

417 The XRD patterns of the Fe₃O₄ and carbon-Fe₃O₄ catalysts are shown in Fig. 6. The peaks
418 indexed as lattice planes (111), (220), (311), (400), (422), (511) and (440) confirm the
419 formation of Fe₃O₄ crystal with a cubic spinel structure (JCPDS card no. 85-1436) [59]. The
420 broader peak shown in the XRD pattern in the range of 20-30° of the carbon-Fe₃O₄ composite
421 material can be related to the lower dimensions of the crystallites and may also suggest the
422 presence of amorphous material.

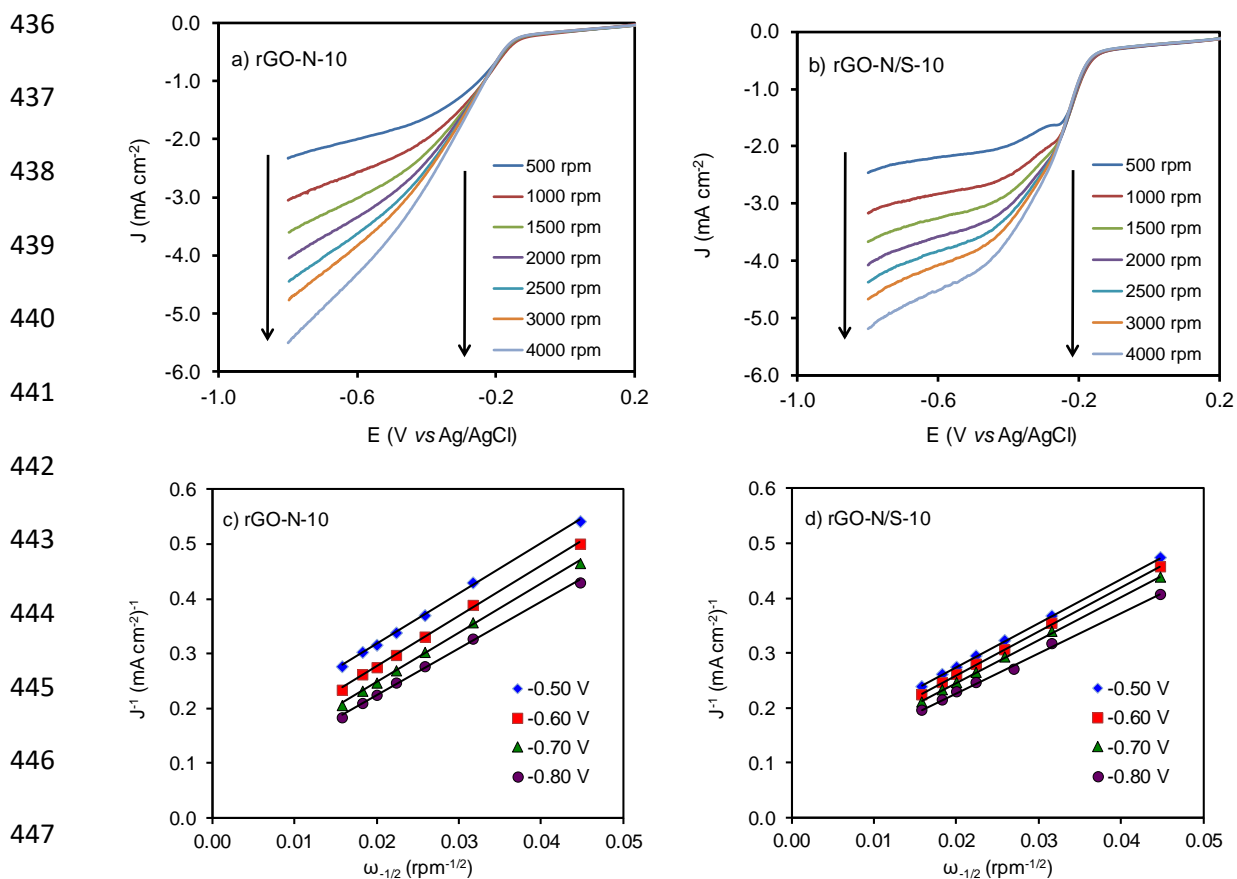


423
424 **Fig. 6.** XRD patterns of Fe₃O₄ and carbon-Fe₃O₄ catalysts.

425 426 3.2. Oxygen reduction reaction: CV and LSV

427 The electrocatalytic ORR activity of the GO and N- and N/S-doped graphene aerogels was
428 firstly assessed by cyclic voltammetry (CV). Fig. S1a-e (supplementary information) depict the
429 electrochemical responses of GO, rGO-N-5, rGO-N-10, rGO-N-20, rGO-N/S-5, rGO-N/S-10
430 and rGO-N/S-20 respectively, under an O₂- or N₂- saturated KOH solution (0.1 M) recorder at
431 a scan of 50 mV s⁻¹ and using a RDE rotation speed of 1000 rpm. The results showed a

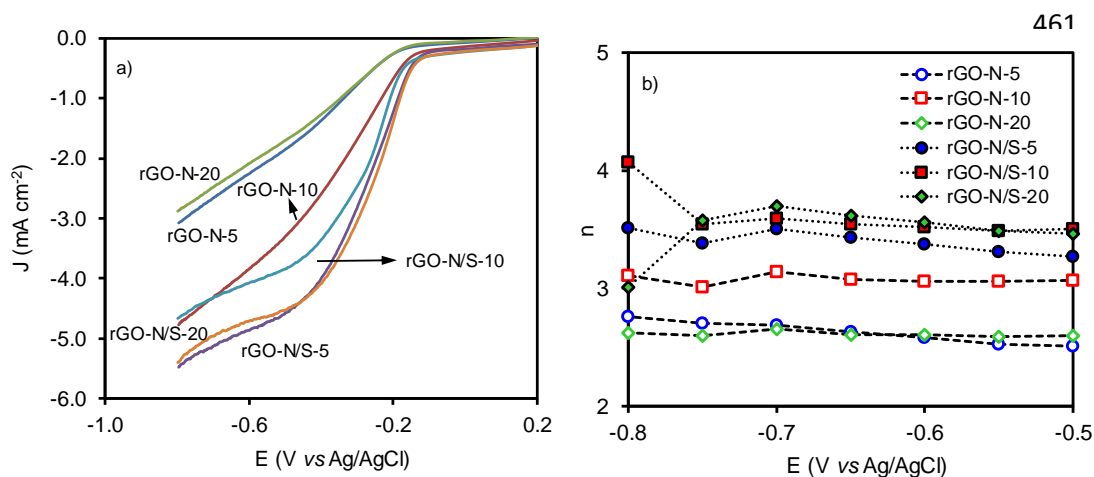
432 significant reduction process under O_2 indicating that all the prepared graphene-based materials
 433 were electroactive to the O_2 reduction. Moreover, no cathodic peak for ORR was shown in N_2 -
 434 saturated electrolyte. In general, the presence of heteroatoms (Fig. S1 b-e) increased the
 435 capacitance significantly in comparison to that obtained for the undoped material (Fig. S1a).



449 **Fig. 7.** a-b) LSV at different rotation speed and c-d) K-L plots at different electrochemical
 450 potentials from -0.5V to -0.8V for rGO-N-10 (a and c) and rGO-N/S-10 (b and d).

451
 452 Among all the heteroatom-doped graphene materials, both rGO-N-5 and rGO-N-20 showed the
 453 lower ORR activity (Figs. S1b and S1d, respectively) whereas all the samples treated with
 454 thiourea (i.e. rGO-N/S-5, rGO-N/S-10 and rGO-N/S-20) exhibited a great performance for
 455 ORR. These results demonstrate the synergistic effect caused by N and S co-doping increasing
 456 the electron density on the surface of the carbon material and promoting the oxygen reduction
 457 reactions as previously reported [22, 60, 61]. These results show that capacitance in carbon

458 materials could be also related to the porous texture. Thus, N/S-doped graphene materials
 459 presented the highest mesopore volume in comparison with the N-doped materials (V_{mes} , Table
 460 1) which are of great importance for the rapid mass transport in electrocatalytic processes [62].



467 **Fig. 8.** a) LSV curves at 3000 rpm and b) variation of n with E vs Ag/AgCl for heteroatom
 468 graphene-based materials.

469

470 LSV measurements at a different rotating speed were also used to examine the ORR activity of
 471 the samples. In particular, Figs. 7a and b depicts the LSV curves for both rGO-N-10 and rGO-
 472 N/S-10, at rotating speeds from 500 to 4000 rpm and in O₂-saturated 0.1 M KOH electrolyte. It
 473 can be seen that the limit current density increases with the increase of rotating speed
 474 correspondingly. In order to obtain the number of electrons transferred at each potential and
 475 evaluate the type of the ORR mechanism (i.e. 4 or 2-electrons), the LSV curves at different
 476 rotation speeds were adjusted to Koutecky-Levich equation (Figs. 7c and d for rGO-N-10 and
 477 rGO-N/S-10, respectively). The corresponding K-L plots exhibit a good parallel linear
 478 relationship from -0.50 to -0.80 V, suggesting the consistent electron transfer number of the
 479 ORR at different electrode potentials. LSV curves at a rotation speed of 3000 rpm for all the
 480 heteroatom doped materials are shown in Fig. 8a. The parameters determined from the analysis
 481 of LSV curves (values of J_k and n refer to K-L fitting for data at -0.80, -0.50 and -0.55V) are
 482 listed in Table 4. Fig. 8b depicted the variation of the number of electrons transferred with the

483 electrochemical potential for each sample. The results showed that both samples rGO-N/S-5
 484 and rGO-N/S-20 exhibited the better performance toward ORR with the higher values of current
 485 density (respectively, 20.933 and 17.298 mA cm⁻² at -0.80 V, Table 4).

486 It is observed that the electron transfer number of the rGO-N/S-10 material is calculated to be
 487 ca. 4.0 at -0.8V, indicating that this material catalysed ORR mainly through a four-electrons
 488 pathway resulting a negligible hydrogen peroxide intermediate generated. Moreover, among
 489 the N/S-heteroatom graphene samples, ORR starts at lower potential using both rGO-N/S-10
 490 and rGO-N/S-20 (i.e. 0.834 and 0.810V, respectively, E⁰_{onset} versus Ag/AgCl). The improved
 491 ORR activity of N/S-doped graphene materials in comparison with N-doped graphene materials
 492 can be ascribed to the enhanced synergetic effect between N and S dopants which can alter the
 493 electronic structure of the sp² carbon atoms as previously reported as well as the favourable
 494 texture contribution from the mesopores volume [22, 61].

495
 496 **Table 4.** Parameters obtained from the analysis of LSV curves for data at -0.8, -0.55 and -0.5
 497 V.

Sample	E ⁰ _{onset}	J _k	n	J _k	n	J _k	n
	(-0.80) (V)	(-0.80) (mA cm ⁻²)	(-0.80)	(-0.55) (mA cm ⁻²)	(-0.55)	(-0.50) (mA cm ⁻²)	(-0.50)
rGO-N-5	-0.206 (0.771) *	6.808	2.76	3.487	2.52	2.943	2.50
rGO-N-10	-0.162 (0.815) *	18.324	3.11	6.068	3.05	7.455	3.07
rGO-N-20	-0.178 (0.799) *	6.245	2.62	3.044	2.59	2.583	2.60
rGO-N/S-5	-0.180 (0.797) *	20.933	3.52	17.319	3.31	15.870	3.27
rGO-N/S-10	-0.143 (0.834) *	13.055	4.05	9.600	3.48	8.870	3.50
rGO-N/S-20	-0.167 (0.810) *	17.298	3.00	14.292	3.49	13.500	3.47
rGO	--	--	--	1.02	1.08	1.10	1.15
GO	--	--	--	0.320	2.45	0.272	3.70

498 *In parentheses, potentials versus the reference hydrogen electrode (RHE) obtained from the conversion of the
 499 measured potentials versus Ag/AgCl reference electrode using the Nernst equation.

500

501 On the other hand, rGO-N-5 and rGO-N-20 exhibited electron transfer number ranges from
502 2.76 to 2.50 and 2.62 to 2.60, respectively, depending on the potential (Table 3). It is well
503 known that the nitrogen content had a significant effect on ORR catalytic activity. Both
504 pyridinic N and graphite N seems to be the main active sites to produce H₂O₂ in 2e⁻ pathway
505 following Eq. (8), as previously reported [18].



507 In general, ORR performance of heteroatom-doped carbon materials is mainly determined by
508 (i) the chemical composition and bonding configurations of the doped atoms and (ii) the specific
509 surface area and porous texture. In our case, N/S-doped graphene materials showed better
510 performance for ORR leading a 4 e⁻ pathway and also a high current density. On the other hand,
511 the introduction of N functionalities in graphene materials favoured the 2e⁻ ORR process with
512 the subsequence H₂O₂ generation. The better catalytic performance of N/S- doped materials
513 might be ascribed to the role of N/S doping which form C-N and C-S bonds (Fig. 5) and could
514 alter the electronic structure of the *sp*² carbon atoms as previously reported [63]. Indeed,
515 positively charged carbon atoms can be generated, due to the electron-accepting nature of
516 nitrogen atoms with respect to carbon atoms. These atoms can act as active sites to adsorb O₂
517 molecules and weaken O-O bonds [22]. Regarding S dopant, the role of S could be related to
518 the mismatch of the farthest orbitals between sulphur and carbon, resulting in the generation
519 of positive charge on the sulphur atoms, which can act as electrocatalytic centers for ORR.
520 Apart from the role of electronic structure, the best ORR performance of N/S-doped graphene
521 materials could be also ascribed to the higher mesoporous structure of these materials compared
522 to N-doped graphene samples (Table 1) which can facilitate the transport of both oxygen and
523 electrolyte, lowering the resistance of mass transport [62].

524 The obtained results indicated that the highest electrogeneration of H₂O₂ at cathode by ORR
525 can be found for both rGO-N-5 and rGO-N-20 catalysts. For this reason, these materials were
526 selected for the subsequent electro-Fenton (EF) experiments. Moreover, a material with mainly

527 4 e⁻ pathway (i.e. rGO-N/S-10) was also studied in the EF experiments in order to compare the
528 results with that of both rGO-N-5 and rGO-N-20 samples.

529

530 *3.3. Adsorption equilibrium studies*

531 Preliminary experiments were performed to establish the adsorption-desorption equilibrium of
532 ACE at room temperature on the selected materials for EF process, i.e. rGO-N-5, rGO-N-20
533 and rGO-N/S-10. Fig. S2 (supplementary information) depicts adsorption isotherms of ACE
534 from distilled water at 298 K on rGO-N-5, rGO-N-20 and rGO-N/S-10 materials. Adsorption
535 isotherms were fitted to Langmuir and Freundlich equations and the results obtained are listed
536 in Table S1 (supplementary information). In general, the linear regression coefficient values
537 show that the Langmuir equation fitted better the adsorption isotherms than the Freundlich one
538 (Table S1). Langmuir equation enables to know the adsorption capacity of the adsorbents and
539 the B constant from which the adsorption thermodynamic parameters can be obtained. The
540 maximum adsorption capacity (X_m) was 116, 120 and 88 mg g⁻¹ for rGO-N-5, rGO-N-20 and
541 rGO-N/S-10, respectively. These results are in accordance with the textural characteristics of
542 the materials obtained, thus, both rGO-N-5, rGO-N-20 catalysts exhibited higher values of
543 surface area and micropore volume (i.e. S_{BET} of 583 and 592 m² g⁻¹ and W_0 of 0.228 and 0.229
544 cm³ g⁻¹, respectively) in comparison with that obtained for rGO-N/S-10 (374 m² g⁻¹ and 0.143
545 cm³ g⁻¹, respectively). Regarding the B constant, the highest value obtained for rGO-N-20 (i.e.
546 0.23 L mg⁻¹) seems to be related with the strongest adsorbent-adsorbate interactions on the rGO-
547 N-20 sample.

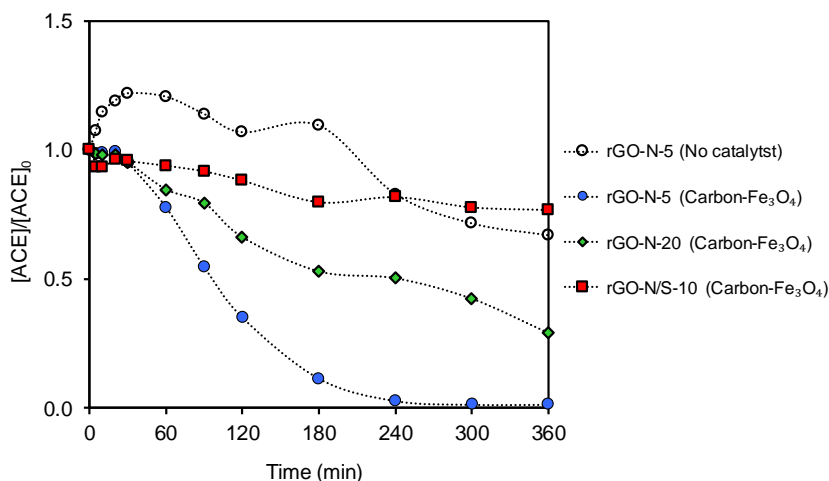
548

549 *3.4. Electro-Fenton*

550 The electro-Fenton (EF) experiments were carried out using the saturated selected heteroatom
551 graphene-based materials, i.e. rGO-N-5, rGO-N-20 and rGO-N/S-10. These catalysts were
552 tested in the degradation of 10 mg L⁻¹ of ACE by EF process at -0.55 V for 360 min. The

553 experiments were performed using carbon-Fe₃O₄ as catalysts to activate the generated H₂O₂ for
554 the degradation of the contaminant (Fig. 9). A blank experiment for the rGO-N-5 sample was
555 investigated in order to quantify the amount of ACE degraded in the absence of magnetic
556 samples. Table 5 summarizes the TOC conversion, ACE conversion and leaching of iron
557 species at the end of the experiments.

558 It well known that the current density is responsible for the reduction of oxygen resulting the
559 formation of hydrogen peroxide at the cathode. Thus, it was observed that for current intensity
560 values under 3.0 mA cm⁻², obtained with potential below -0.5 V (Table 4), a negligible ACE
561 degradation was achieved (results not shown). These results seem to be related with the
562 inefficient oxygen reduction due to the lower energy supplied to the system as previously
563 reported [28]. For the EF experiments, potential value of -0.55 V was selected for the EF
564 reaction, according to the obtained values of J_k , n (Table 4) and the respective ACE degradation
565 (Table 5). For the materials in which the electron transfer number is ca. 2 at -0.55V (i.e. 2.52
566 and 2.59 for rGO-N-5 and rGO-N-20, respectively) and using carbon-Fe₃O₄ as catalyst, the EF
567 results showed that rGO-N-5 promote the highest ACE conversion, reaching almost complete
568 removal in 240 min whereas rGO-N-20 exhibited a 71% of ACE conversion (Table 2).
569 However, a different tendency was observed for the ACE mineralization; i.e. after 360 min;
570 rGO-N-5 (+ Carbon-Fe₃O₄) produced a TOC reduction of 45.5% while rGO-N-20 leads to a
571 TOC reduction of 51.6% (Table 5). These results could to be related with the the larger amount
572 of iron leached obtained for rGO-N-20 compared to that obtained for rGO-N-5 (respectively
573 0.33 and 0.24 mgL⁻¹). On the other hand, and as expected, rGO-N/S-10 showed the lowest
574 performance (23 and 13.1% of ACE conversion and mineralization, respectively) due to the
575 mainly 4-electron pathway for ORR that implies a much lower generation of H₂O₂ (Table 4) in
576 comparison with that obtained for rGO-N-5 or rGO-N-20 samples.



577

578 **Fig. 9.** Normalized ACE concentration ($[ACE]/[ACE]_0$) as function of time for electro-Fenton
 579 experiments performed with the saturated samples rGO-N-5, rGO-N-20 and rGO-N/S-10 and
 580 using carbon-Fe₃O₄ as catalyst at potential of -0.55V. A non-catalytic blank experiment for
 581 rGO-N-5 without catalyst is also shown for comparison.

582

583 The influence of pH on the EF degradation of ACE at natural solution pH (i.e. 5.7) is shown in
 584 Fig. S4 (supplementary information). As expected, the efficiency of the EF process decreases
 585 at higher initial solution pH. This can be explained by the precipitation of iron species as ferric
 586 hydroxides as well as the decomposition of generated H₂O₂ into water and O₂, thus hindering
 587 the production of HO[•], as previously reported [31]. Nevertheless, it is of special interest to note
 588 that ACE degradation of approximately 70% was obtained using rGO-N-5 at neutral reaction
 589 pH (Fig. S4). More studies will be further required to improve the catalyst stability at natural
 590 solution pH.

591 A blank experiment was also investigated in order to quantify the amount of ACE degraded
 592 under non-catalytic conditions (rGO-N-5 no catalyst). The results showed that although a
 593 desorption of ACE was occurring during the first 180 minutes, an ACE conversion of 33% with
 594 a TOC reduction of 5.0% was achieved at the end of the experiments, i.e. 360 min (Table 5).
 595 These results could be due to the formation of HO[•] radicals in the absence of catalyst. In fact,
 596 it has been reported that the presence of N-functionalities (such as pyridinic N and graphitic)

597 could catalyze H₂O₂ to the production of radicals HO• [18]. However, the catalysis of H₂O₂ to
 598 HO• was much slower than that of H₂O₂ generation, which lead to the accumulation of H₂O₂.
 599 For that reason, the use of an active catalyst, i.e. carbon-magnetite is crucial to accelerate the
 600 generation of radicals, favoring the conversion and degradation of ACE (Table 5).
 601 Heteroatom graphene-based materials could promote the 2e⁻ ORR process to electrochemically
 602 generate H₂O₂ and the presence of carbon-Fe₃O₄ could efficiently catalyze H₂O₂ to produce
 603 HO• radical for ACE degradation. In literature, the reaction mechanism seems to be mostly
 604 motivated by HO• radicals attack during the EF process, although the formation of other reactive
 605 species such as radical superoxide anion (O₂^{•-}) can assume a representative role in the EF
 606 mechanism [18] following Eq. (9). After the generation of radicals, ACE might be attacked and
 607 degraded into different aromatic by-products such as benzoic acid, benzaldehyde and
 608 acetamide, among others, as well as aliphatic organic acids such as fumaric, oxalic, acetic,
 609 formic and maleic acids, as previously reported [64]. Even so, more studies will be further
 610 required, perhaps using more advanced characterization techniques for exact identification of
 611 all reaction intermediates of ACE.



613 On the other hand, post-reaction analysis of the rGO-N-5 and rGO-N-20 (EF) samples by XPS
 614 (results not shown) indicated that nitrogen functionalities were stable and there are not
 615 significant changes of the N groups after ACE oxidation by EF.

616

617 **Table 5.** ACE conversion ($X_{ACE,360min}$), total organic carbon conversion ($X_{TOC,360min}$) and
 618 leaching of iron species ($C_{Fe-leached}$) in EF experiments at -0.55V after 360 min.

Sample	$X_{ACE,360min}$ (%)	$X_{TOC,360min}$ (%)	$C_{Fe-leached}$ (mg L ⁻¹)
Blank (rGO-N-5 no catalyst)	33	5.0	--
rGO-N-5 (+carbon-Fe ₃ O ₄)	100	45.5	0.24
rGO-N-20 (+carbon-Fe ₃ O ₄)	71	51.6	0.33
rGO-N/S-10 (+carbon-Fe ₃ O ₄)	23	13.1	0.15

619

620 Regarding the Fe leached at 360min (Table 5), it is noteworthy that all the catalysts tested
621 present values much lower than the maximum dissolved iron concentration allowed in
622 European Union standards for discharge of treated waters (i.e. 2.0 mg L^{-1}). The concentration
623 of leaching of iron species was 0.24, 0.33 and 0.15 mg L^{-1} for rGO-N-5 (+carbon- Fe_3O_4), rGO-
624 N-20 (+carbon- Fe_3O_4) and rGO-N/S-10 (+carbon- Fe_3O_4), respectively. The lower amount of
625 iron leached obtained for all the samples indicate that the Fenton reaction is taking place
626 heterogeneously between iron supported on carbon and the generated H_2O_2 in solution,
627 confirming the higher stability of the catalysts. Besides the high stability of the carbon catalyst,
628 they have an additional advantage of the magnetically recoverable catalysts.

629

630 **4. Conclusions**

631

632 A series of N- and N/S-doped graphene aerogels were prepared by simple hydrothermal method
633 and freezing-drying process using urea or thiourea as N or N/S source, respectively.
634 Electrochemical characterization was carried out to assess the electro-catalytic performance and
635 the number of electrons transferred of each sample for ORR. The N-functionalities introduced
636 in the graphene-based material remarkably influence the catalytic performance in both
637 processes. In general, N/S-doped graphene aerogels show better performance for ORR leading
638 a four- electron pathway and also high current densities. The improved catalytic performance
639 can be attributed to the enhanced synergetic effect between N and S dopants. The introduction
640 of N functionalities in GO favour the $2e^-$ ORR process for H_2O_2 generation. Heterogeneous
641 electro-Fenton process for the degradation of acetaminophen using both N-graphene aerogels
642 and magnetically recoverable catalyst based on carbon-magnetite, allowed a contaminant
643 removal of ca. 100%, ca. 45% of mineralization and also a low iron loss by leaching (ca. 0.24

644 mg L⁻¹), conforming with the European Union directives for discharge of treated waters. The
645 selected magnetic carbon catalyst also exhibits good stability.

646
647

648 **Acknowledgments**

649 LMPM acknowledges the Spanish Ministry of Economy and Competitiveness (MINECO) and
650 the European Social Fund for a Ramon y Cajal research contract (RYC-2016-19347). NFS
651 acknowledges the National Youth Guarantee System (SNGJ) and European Social Fund (ESF)
652 for the training contract. DEVM acknowledges CONACyT, Mexico, through grant No.:
653 297739. This research was supported by the RNM-172 research group and by the project
654 RTI2018-099224-B-I00 funded by Spanish MINECO.

655

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